

# Journal Pre-proof

## Factors Determining Self-Assembly of Hyaluronan

Karolína Kocourková (Investigation) (Formal analysis) (Visualization) (Writing - original draft), Lenka Musilová (Formal analysis), Petr Smolka (Writing - review and editing), Aleš Mráček (Project administration) (Resources), Martin Humeník (Investigation) (Formal analysis) (Writing - review and editing), Antonín Minařík (Conceptualization) (Methodology) (Supervision) (Writing - review and editing)



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# Factors Determining Self-Assembly of Hyaluronan

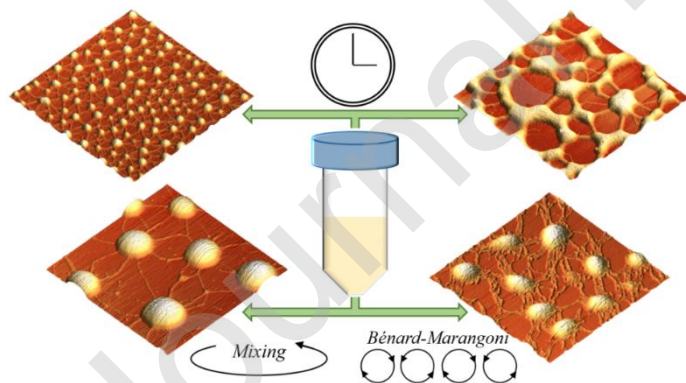
Karolína Kocourková,<sup>†,‡</sup> Lenka Musilová,<sup>†,‡</sup> Petr Smolka,<sup>†,‡</sup> Aleš Mráček,<sup>†,‡</sup> Martin Humeník,<sup>§</sup>  
Antonín Minařík,<sup>\*,†,‡</sup>

<sup>†</sup> Department of Physics and Materials Engineering, Tomas Bata University in Zlín, Vavrečkova  
275, 760 01 Zlín, Czech Republic

<sup>‡</sup> Centre of Polymer Systems, Tomas Bata University in Zlín, Třída Tomáš Bati 5678, 760 01  
Zlín, Czech Republic

<sup>§</sup> Department of Biomaterials, Faculty of Engineering Science, Universität Bayreuth, Prof.-  
Rüdiger-Bormann.Str. 1, 95447 Bayreuth, Germany

## Graphical abstract



## Highlights

- The network-forming and aggregation properties of hyaluronan were studied.

- Effect of dissolution conditions and storage time was proved.
- Specific changes in mesh density and aggregation rate were observed.
- Structures on solid surface reflects properties of hyaluronan in solution.

## ABSTRACT

The conditions determining network-forming and aggregation properties of hyaluronan on the mica surface were studied. The hyaluronan was deposited on the surface from aqueous and saline solutions and attached by a bivalent cation. The morphology of the immobilized assemblies was characterized by atomic force microscopy. The experimental results show that the morphology and size of the aggregates as well as the density of the interconnecting fibrillar network, both made of hyaluronan, at the liquid-solid phase interface are determined not only by its molecular weight or concentration in solution, but also by the dissolution conditions and storage time.

These findings extend the current state of knowledge about the conformational variability of this biologically important polymer. Understanding the conformational variability is of great importance, as it governs the physiological functions of hyaluronan, as well as its processability and formulations. That in turn determines its usability in different pharmacological and biomaterial applications.

### *Keywords*

Hyaluronan, atomic force microscopy, polysaccharides, network, aggregates

## 1. Introduction

Hyaluronic acid, or rather its sodium salt – hyaluronan (HU), has a number of important biological functions, which are closely related to its molecular weight, concentration and spatial arrangement of the polymer chains (Dicker et al., 2014; Evanko, Parks & Wight, 2016). Due to its natural occurrence in the human body, it is used in a wide range of pharmaceutical and biomedical applications (Dosio, Arpicco, Stella & Fattal, 2016; Highley, Prestwich & Burdick, 2016). Despite the considerable amount of work devoted to this polysaccharide, new studies are constantly expanding the knowledge-base on this macromolecular system (Fallacara, Baldini, Manfredini & Vertuani, 2018; Grossutti & Dutcher, 2016; Mondschein, Kanitkar, Williams, Verbridge & Long, 2017).

The specific properties and bioactivity of HU are closely related to the spatial arrangement of the macromolecule. The conformational configuration of HU in solution varies radically over a wide range of concentrations from a random coil form with a significant degree of flexibility to semi-organized polymer networks (Day & Sheehan, 2001). These networks are highly complex in a biological environment (Heinegård, 2009). They affect the transport of water, limit the movement of pathogens (Xu, Jha, Harrington, Farach-Carson & Jia, 2012), efficiently absorb vibrations in extracellular environment, effectively control the diffusion of proteins (Dicker et al., 2014) and are among the components responsible for the mechanical properties of soft tissues (Scott, Cummings, Brass & Chen, 1991). The formation and stability of the network structure in tissues is determined by high concentration of HU. For example, human vitreous humor contains HU of molecular weight  $2-4 \times 10^6$  Da in concentration 65-400  $\mu\text{g/ml}$  (Bishop, 2000).

In contrast, HU networks under *in vitro* conditions are much simpler. They consist of randomly interconnected HU fibres with unstable mechanical properties (Scott & Heatley, 1999;

Xu et al., 2012). The impact of this can be observed, for example, in the treatment of osteoarthritis by injecting HU as lubricant, where the viscosity of the HU solution decreases dramatically under shear stress. This problem can be overcome by chemical treatment or formulations leading to morphological changes of the HU networks (Zheng et al., 2019). It is known that the preparation of HU-based hydrogels and scaffolds under *in vitro* conditions very often depends on the functionalization of polymer chains (Deng et al., 2017; Foglarová et al., 2016; Nimmo, Owen & Shoichet, 2011; Van Vlierberghe, Dubruel & Schacht, 2011).

In diluted solutions, HU chains occur in the form of random coils that form a temporary polymer network, while at high concentrations the network density increases and the macromolecules cluster into aggregates (Lapčik, Lapčik, De Smedt, Demeester & Chabreček, 1998; Matteinie, 2009; Scott et al., 1991). The driving forces of aggregation are hydrophobic interactions and hydrogen bonds compensating the electrostatic resistance between polyanionic charges (Gribbon, Heng & Hardingham, 1999; Rinaudo, 2006; Scott, 1992). Further, the properties of the polymer network are strongly dependent on the molecular weight of HU (Shen, Chaudouet, Ji & Picart, 2011; Wieland et al., 2016). Longer chains show more extensive interweaving and higher number of interchain interactions. The resulting network is therefore more coherent and larger (Scott et al., 1991). The formation of HU network and the conformation changes are also conditioned by the pH of the solution and by the concentration, type and valence of the ions in the solution (Giubertoni et al., 2019; Gřundělová, Mráček, Kašpárková, Minařík & Smolka, 2013; Mráček et al., 2015; Zellermann, Bergmann & Mayer, 2013).

Different characteristics and HU networks were analyzed by rotary shadowing electron microscopy (Scott et al., 1991), X-ray crystallography (Hargittai & Hargittai, 2008), NMR spectroscopy (Almond, DeAngelis & Blundell, 2006; Cowman, Hittner & Feder-Davis, 1996;

Nestor & Sandström, 2017; Scott & Heatley, 1999), light scattering (Kim, Woo, Park, Hwang & Moon, 2015), tensiometry (Musilová, Kašpárková, Mráček, Minařík & Minařík, 2019), chromatography (Shanmuga Doss, Bhatt & Jayaraman, 2017), molecular dynamics (Cowman & Matsuoka, 2005; Hargittai & Hargittai, 2008; Kutálková, Hrnčířik, Witasek & Ingr, 2020) and atomic force microscopy (AFM) (Cowman et al., 2005; Cowman, Li & Balazs, 1998; Giannotti, Rinaudo & Vancso, 2007; Gunning, Morris, Al-Assaf & Phillips, 1996; Jacoboni, Valdrè, Mori, Quaglino & Pasquali-Ronchetti, 1999; Spagnoli et al., 2005; Zellermann et al., 2013). It has been proven, that despite the effect of the surface and adsorbed water described by Cowman (Cowman et al., 2005), AFM can be used to study the aggregation and networking ability of HU with respect to the process conditions of solution preparation. For that reason, the AFM microscopy has been chosen as a dominant experimental technique in this work.

Herein, we show that the self-assembly of HU is controlled by the combination of chemical factors such as molecular weight ( $MW$ ) and concentration of the biopolymer with processing factors such as temperature, dissolution method and storage time of the polymer solution. All these variables affect the physiologically important characteristics of the biopolymer system. The present study aims at establishing the link between processing conditions of solution preparation and aggregation behavior on interfaces, which is required for understanding the HU-based material properties such as coatings and films.

## **2. Materials and methods**

### **2.1. Chemicals**

Bacterial sodium salt of hyaluronic acid of various molecular weight in pharmaceutical purity was purchased from Contipro a.s. (Czech Republic). The weight average molecular weight ( $M_w$ ) of HU was in a range from  $\sim 190$  kDa to  $\sim 1000$  kDa as characterized by the AF4-MALS

chromatography. As a dissolving medium, 0.9% sodium chloride (NaCl) solution (physiological solutions) acquired from Sigma-Aldrich and the ultra-pure water of resistance 18.2 M $\Omega$  was used. Magnesium chloride (MgCl<sub>2</sub>) for the mica surface modification was also purchased from Sigma-Aldrich, as well as Tris/HCl buffer. Mica, serving as a substrate for HU to AFM imaging, was obtained from SPI SUPPLIES in quality V-4.

## 2.2. Molecular weight analysis

The asymmetrical flow field-flow fractionation (AF4) was performed using the large channel (LC model) with trapezoidal design from Wyatt Technology Europe GmbH (Dernbach, Germany) equipped with the regenerated cellulose membrane (MWCO 30 kDa) from Merck Millipore (Darmstadt, Germany) and 350  $\mu$ m spacer.

The HU samples were dissolved in 50 mM Tris/HCl buffer, pH 8.0 (the carrier solution) at 1 mg/mL at RT for 16 h. Injection volume was 180  $\mu$ l for each sample. The carrier solution was filtered through a nitrocellulose filter (pore size = 0.2  $\mu$ m) from Millipore. After that, it was delivered by an Agilent 1100 Series HPLC system (Agilent, Germany).. The system was additionally connected to a multi-angle light scattering detector DAWN EOS (Wyatt, Germany) and a refractive index detector Shodex RI-71 (Shodex, Germany). Molecular masses and root mean square (rms) radii from light scattering signals were calculated using the ASTRA 6 software (Wyatt, Germany).

## 2.3. Solution preparation

HU aqueous or physiological solutions with a concentration of 0.5 wt.% were prepared by different dissolving ways of powder HU. The first type of solution was prepared on the laboratory orbital shaker at room temperature for 24 or 48 h. The second way was realized on the magnetic stirrer at 45 °C for 24 h and the last type of process involved dissolving in the



temperature field flow chamber (TFFC) in temperature gradient for 24 h. Temperature gradient was generated by the temperature difference between the top cooling plate with 20 °C and the bottom heating plate with 50 °C, as described in our previous works (Minařík et al., 2017). These solutions were stored at 4 °C and diluted to the concentration of 0.1 wt. % or 0.01 wt. % prior to the preparation of the sample for AFM.

#### **2.4. Atomic Force Microscopy**

HU macromolecules deposited on the mica surface were imaged by the AFM. The freshly cleaved mica of area 1 cm<sup>2</sup> was treated by 0.1 M MgCl<sub>2</sub>. This solution was deposited on the mica surface in volume of 50 µl and dried by air flow after 60 s of incubation at room temperature. Then, 50 µl of 0.1 or 0.01 wt. % HU solution was deposited on the dry modified mica surface and incubated for 120 s at room temperature. The excess solution was then gently dried by air flow.

Unless otherwise stated, all process parameters for the deposition of the solution on the mica surface were kept constant. The samples were measured by NTEGRA Prima (NT-MDT) by tapping mode in air. The silicon probe of force constant (1.45-15.1) N/m, type NSG01 (TipsNano) was used. The sample area was scanned in rate of 0.7 Hz in 512x512 pixel format.

#### **2.5. Image analysis**

Images were processed using the Gwyddion – Free SPM data analysis software, version 2.55 (D. Nečas, P. Klapetek, Czech Metrology Institute, Czech Republic). The area coverage was analyzed by the ImageJ software, version 1.5 (W. Rasband, National Institutes of Health, United States), the aggregates and network surface covering was obtained and averaged from 5 areas, 1 µm<sup>2</sup> each. Herein, the image analysis is presented for the most significant statistical data. Complete numerical analysis for all images can be found in Supplementary information.

### 3. Results and discussion

Field-flow fractionation (FFF) is based on an open-channel technique for fractionation of molecular mixtures, which allows avoiding many problems in polymer analysis related to the stationary phase such as unspecific adsorption and extensive shear forces applied on large macromolecules. A FFF sub-technique, asymmetric flow field-flow fractionation (AF4) allows a fractionation of complex polymer systems according to molecular size by application of longitudinal and cross-oriented flows in the channel (Kim et al., 2015; Malik & Pasch, 2016; Shin, Hwang, Cho & Moon, 2007). The hyaluronan used in the study was characterized first using AF4 connected to MALS detectors (Wyatt, 1997) in order to obtain molecular masses, polydispersity and shape factors of the polysaccharide chains in solutions (Table 1).

**Table 1**

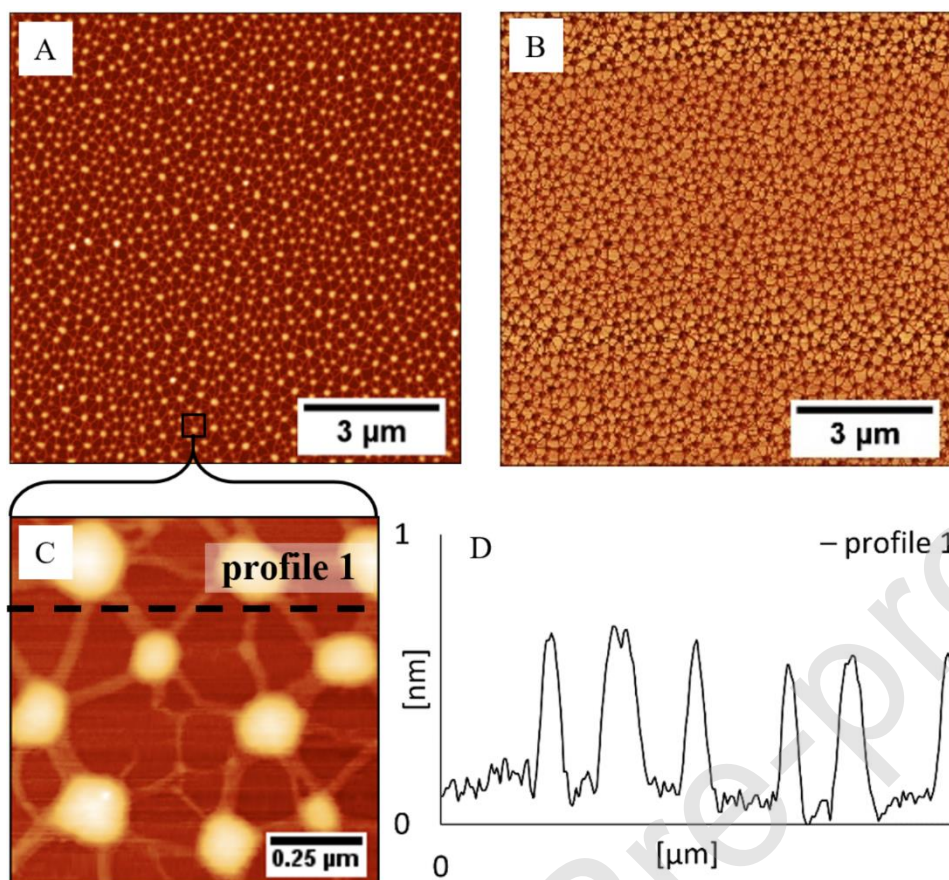
Molecular weights (number and weight average  $M_n$  and  $M_w$ ), polydispersity and RMS radii  $R_n$  and  $R_w$  (corresponding to  $M_n$  and  $M_w$ ) of different hyaluronans calculated from AF4-MALS analysis.

Sample	$M_n$ (kDa)	$M_w$ (kDa)	Polydispersity ( $M_w/M_n$ )	$R_n$ (nm)	$R_w$ (nm)	Conformation plot slope
HU190	161.8	186.2	1.151	33.2	37.3	0.49
HU470	377	473.3	1.255	58.2	63.2	0.23
HU720	668.2	723	1.082	93.4	98.7	0.60
HU1000	987.8	1006.5	1.019	128.1	130	0.58

Evaluation of conformations slopes ( $< 0.33$  for sphere,  $0.5\sim 0.6$  for linear random coil, and  $>1$  for rigid rod) revealed rather compact conformations in the used low ionic strength buffer systems.

AFM imaging of HU macromolecules has been presented in a number of studies demonstrating that their conformation is closely related to the properties of the substrate and the sample preparation methodology. While rinsing the excess solution from the carrier substrate with water leads to the attachment of the isolated chains on the surface by the process called “molecular combing”, by omitting this step it is possible to obtain aggregated clusters of macromolecules interconnected by bundles of chains forming network (Cowman et al., 2005; Spagnoli et al., 2005). In the present study, we used this texture to present a connection between solution history and conformation of solid-state polymer macromolecules. Using the AFM, the influence of different approaches to solution preparation was found not only on the character of the physical network reflecting the properties of the solution, but also the aggregation ability of the polymer on the surface of the carrier substrate.

Fig. 1 shows a topography image of a physical HU network (HU1000) deposited from a 0.1 wt. % solution homogeneously covering the substrate surface (Fig. 1A) including phase contrast (Fig. 1B). In the detail in Fig. 1C, the fibers forming the network are shown. The fiber height of  $0.75 \pm 0.04$  nm, seen from the profile section in Fig. 1D, corresponds to the diameter of the two polysaccharide chains of HU (Cowman et al., 2005).

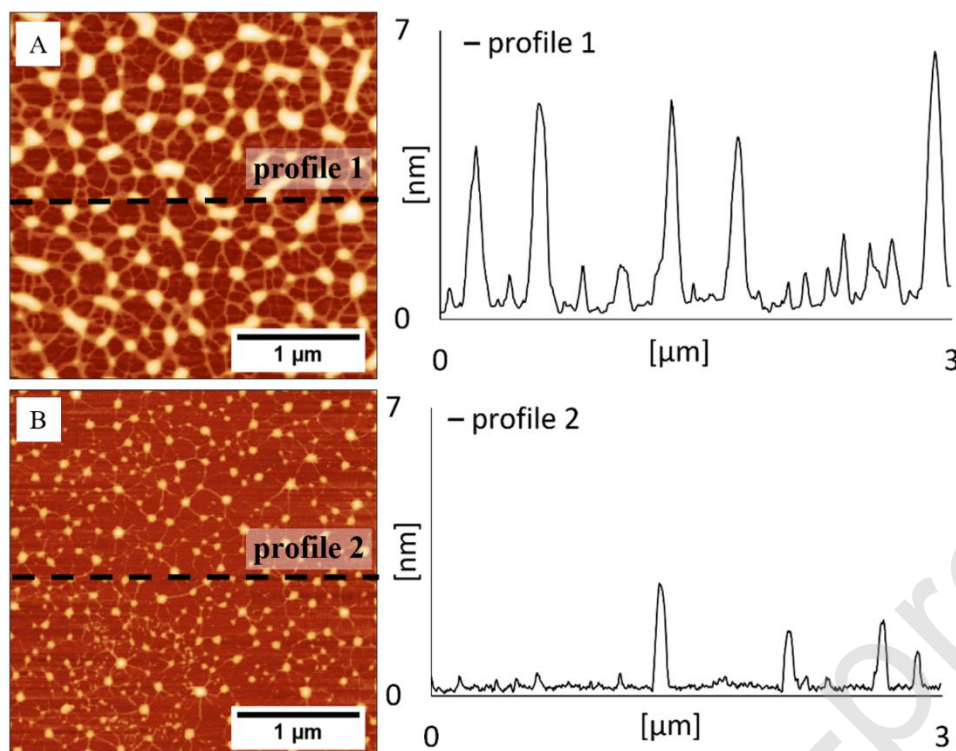


**Fig. 1.** HU network on the mica surface deposited from 0.1 wt. % aqueous solution:

A) topography ( $S_z = 12 \text{ nm}$ ) and B) phase contrast ( $S_z = 12^\circ$ ) of the HU1000 network, C) detail of HU chains interconnecting the isolated aggregates ( $S_z = 10 \text{ nm}$ ) with D) profile section.  $S_z$  means the maximum height of the analyzed area.

### 3.1. Concentration of deposited solution

According to the literature, the intermediate polymeric network is formed in solution even at very low concentrations below 0.01 wt. % (Spagnoli et al., 2005), where aggregate clusters are partly formed in solution and are completed when HU is deposited on the surface during drying (Cowman et al., 2005; Spagnoli et al., 2005). When diluting the initial 0.5 wt. % solution before deposition on the surface, significant differences in the size and frequency of the aggregates are observed, as seen from the Fig. 2A (0.1 wt. %) and 2B (0.01 wt. %).

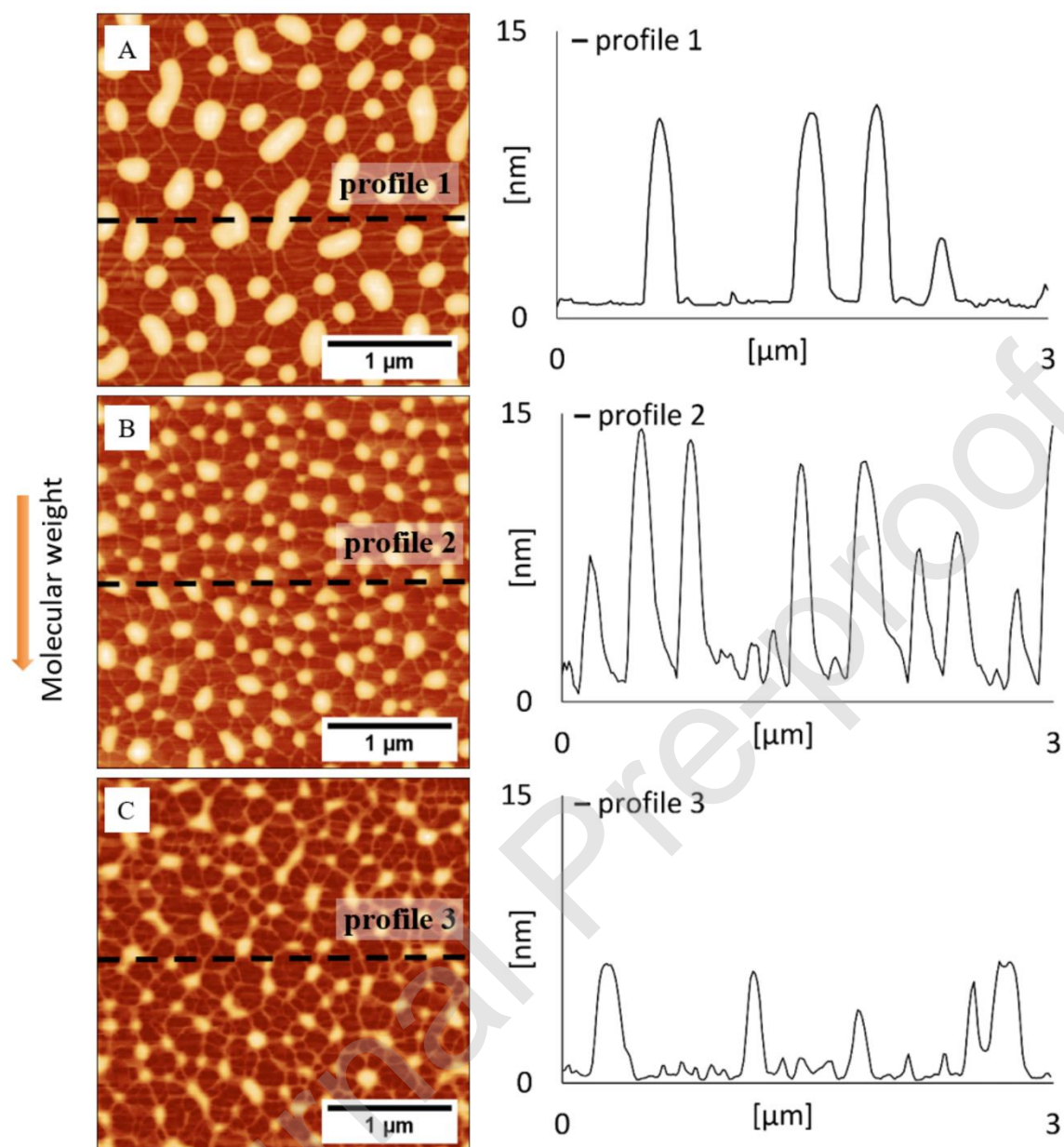


**Fig. 2.** AFM topography images of networks from solution of HU1000 dissolved at concentration of 0.5 wt. % on the orbital shaker at 24 °C for 24h deposited at concentrations of A) 0.1 wt. % ( $S_z = 8 \text{ nm}$ ) and B) 0.01 wt. % ( $S_z = 4 \text{ nm}$ ). On the right, the cross sections are shown.  $S_z$  means the maximum height of the analyzed area.

It is apparent from the profile sections next to the topography images (Fig. 2) that the height of the aggregates decreased approximately by half at a lower concentration. HU deposited at a concentration of 0.1 wt. % provides uniform surface coverage in one layer of the polymeric physical network. This observation corresponds to the knowledge of aggregation and netting concentration-dependence of HU in solution and on the phase interface and means that the selected immobilization and imaging of HU on the mica surface reflects directly the state of this macromolecular system in solution (Cowman et al., 2005; Matteini et al., 2009).

### 3.2. Molecular weight

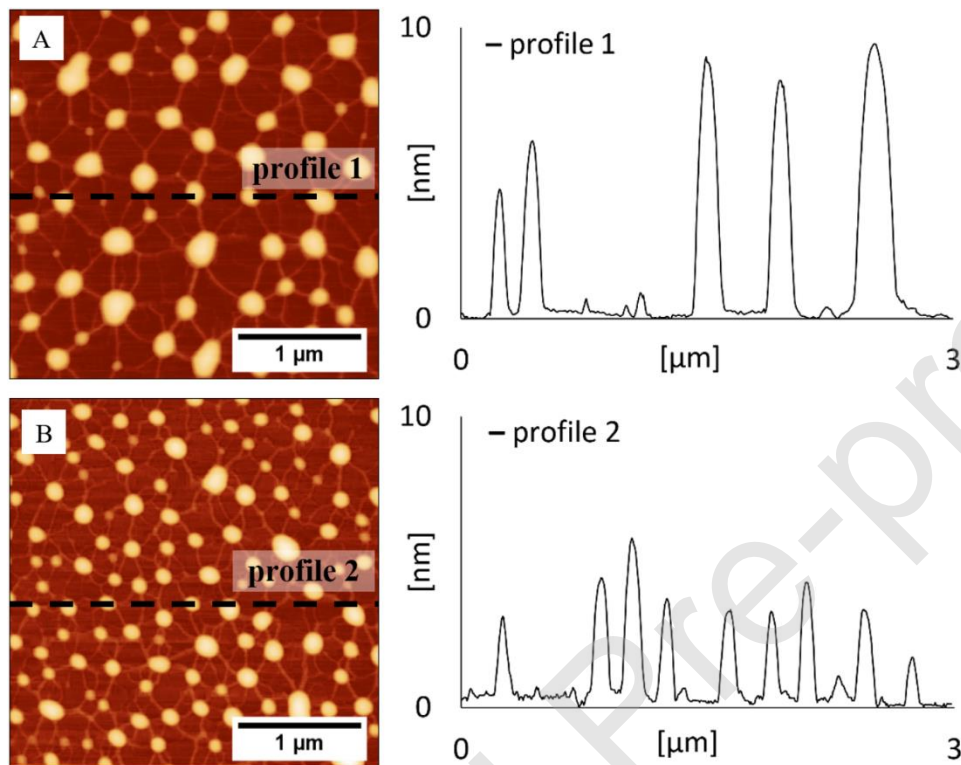
The effect of the molecular weight of the deposited HU on the character of the network is presented in Fig. 3. As can be seen, HU190 with the molecular weight of 190 kDa (Fig. 3A) shows a significantly lower mesh density compared to HU1000 (Fig. 3C). This phenomenon is consistent with the claim that long molecules form more nodes in solution than short molecules at the same concentration (Scott et al., 1991). For lower molecular weight HU, there is less opportunity for intermolecular interactions. For this reason, low  $M_w$  HU occur in solution as separate entities capable of forming organized domains more easily rearranged within individual polymer clusters. The free HU chains can then aggregate into larger clusters. The result is a decreasing number of aggregates and their enlargement.



**Fig. 3.** AFM topography images of networks from HU solutions A) HU190 ( $S_z = 13 \text{ nm}$ ) B) HU470 ( $S_z = 17 \text{ nm}$ ) and C) HU1000 ( $S_z = 14 \text{ nm}$ ) with starting concentration 0.5 wt.% deposited in concentration 0.1 wt. % dissolved in 24 h at 24 °C on the shaking machine.  $S_z$  means the maximum height of the analyzed area. On the right, the cross sections are shown.

### 3.3. The shaking time during dissolving

In case of dissolution of HU on the orbital shaker, one of the variables was the dissolution time. Solutions containing 0.1 % HU720 were prepared on the orbital shaker for 24 h and 48 h.



**Fig. 4.** AFM topography images of networks from solution of HU720 deposited in concentration 0.1 wt. % dissolved on the shaker at 24 °C for A) 24 h ( $S_z = 17 \text{ nm}$ ) and B) 48 h ( $S_z = 11 \text{ nm}$ ).  $S_z$  means the maximum height of the analyzed area. On the right, the cross sections are shown.

The results in Fig. 4 and the image analysis in Table 2 show that the longer mixing time results into doubled number of aggregates and increased number of crosslinks, while the coverage of the area with aggregates and nets remains similar. The results indicate that with prolonged influence of shear forces in the solution the shaking triggers disintegration of aggregates and higher crosslinking of the polymer chains.



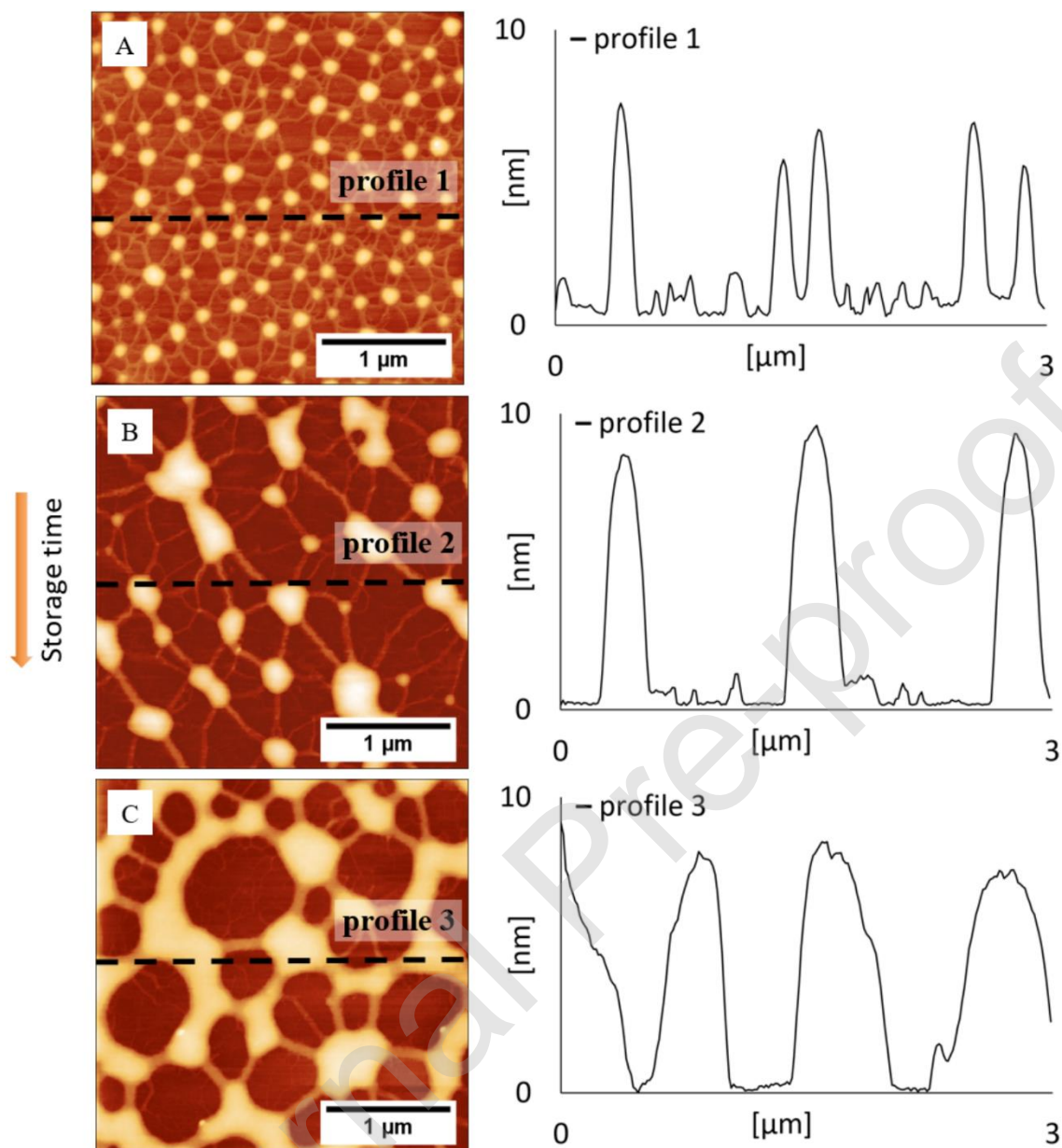
**Table 2**

Comparison of network structures from solutions with different dissolving time, analyzed area 1  $\mu\text{m}^2$

Dissolution time [h]	Aggregates number	Maximum height [nm]	Mean height [nm]	Average projected area [ $10^3 \text{ nm}^2$ ]	Aggregates covering [%]	Network covering [%]	Total covering [%]
24	$9.8 \pm 1.2$	$12.1 \pm 1.2$	$5.2 \pm 0.3$	$22 \pm 2$	$18.0 \pm 0.6$	$12.0 \pm 2.0$	$29.9 \pm 1.5$
48	$17.0 \pm 0.8$	$8.2 \pm 0.4$	$3.9 \pm 0.1$	$9 \pm 1$	$15.5 \pm 0.7$	$15.7 \pm 1.7$	$31.2 \pm 1.4$

### 3.4. Storage

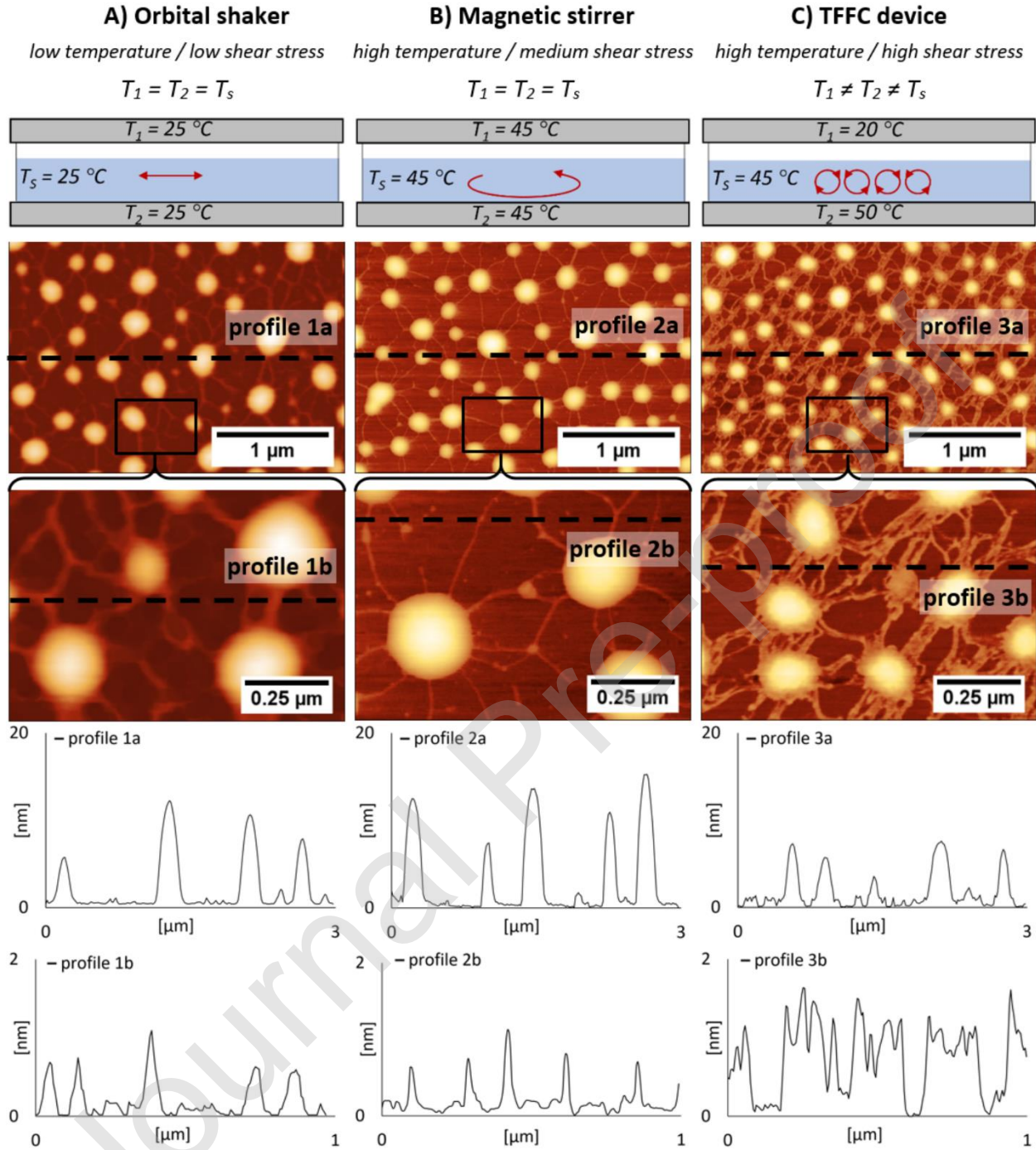
The effect of storage on the formation of surface structures is presented in Fig. 5. HU1000 was dissolved at room temperature for 67 h on the shaking machine and then stored at quiescent condition at  $4^\circ\text{C}$  for 2 -- 4 days. Assuming that the HU chains are not degraded during the storage (Simulescu, Kalina, Mondek & Pekař, 2016), it is likely that the polymer network rearranged in the absence of shear forces since, intrachain interactions are favored over interchain ones. As a result, increasingly spatially distinct clusters are formed.



**Fig. 5.** Effect of the solution storage time on HU1000 network-forming and aggregation properties. AFM sample preparation A) Day 2 ( $S_z = 13 \text{ nm}$ ), B) Day 3 ( $S_z = 15 \text{ nm}$ ), C) Day 4 ( $S_z = 13 \text{ nm}$ ) at  $4^\circ\text{C}$  after dissolution on shaking machine at room temperature for 67 h. HU1000 was deposited from 0.1 wt. % solution.  $S_z$  means the maximum height of the analyzed area. On the right, the cross sections are shown.

### 3.5. Shear influence

From the processing point of view, a method of preparation with different types of solution flow during HU dissolution was followed. Similar results were obtained by stirring the solution during dissolving by orbital shaker at 25 °C and magnetic stirrer at 45 °C as shown in Fig. 6 A and B, respectively. However, a significant difference was noted in case of dissolving HU in the temperature field flow chamber (TFFC, Fig. 6 C). In TFFC, temperature gradients (50/20 °C, middle temperature in solution  $T_s = 45^\circ\text{C}$ ) influence a rearrangement of macromolecules in solution due to generated Bénard-Marangoni convection, as described in our previous work (Minařík et al., 2017). The self-organized fluid flows associated with intense shear stress of macromolecular chains that occurs in the solution during dissolution affects the frequency of interchain interactions and causes a higher cross-linking density, which is evident from the higher frequency of aggregates, or nodes, in the polymer network on the surface. Interestingly, the convective flows in the solution layer triggered multiple number of polymer bundles connecting the individual aggregates.



**Fig. 6.** AFM topography images of networks from solution of HU470 solubilized at concentration of 0.5 wt. % and deposited at a concentration of 0.1 wt. %. A) solubilization on orbital shaker at 25 °C for 24 h ( $S_z = 16\text{ nm}$ ), B) solubilization on magnetic stirrer at 45 °C for 24 h ( $S_z = 21\text{ nm}$ ) and C) solubilization in TFFC at temperature gradient from 20 to 50 °C,  $T_s=45^\circ\text{C}$

( $S_z = 12 \text{ nm}$ ).  $S_z$  means the maximum height of the analyzed area. Cross sections along the dashed lines are shown below the respective AFM images.

Despite the slightly higher number of aggregates in the case of TFFC, due to their lower diameter, the area coverage with aggregates for all approaches was  $\sim 20 \%$ . However, the surface coverage by the fibrillar networks differs significantly. In the case of TFFC, the net coverage is almost twice as high as with magnetic stirrer, see Table 3.

**Table 3**

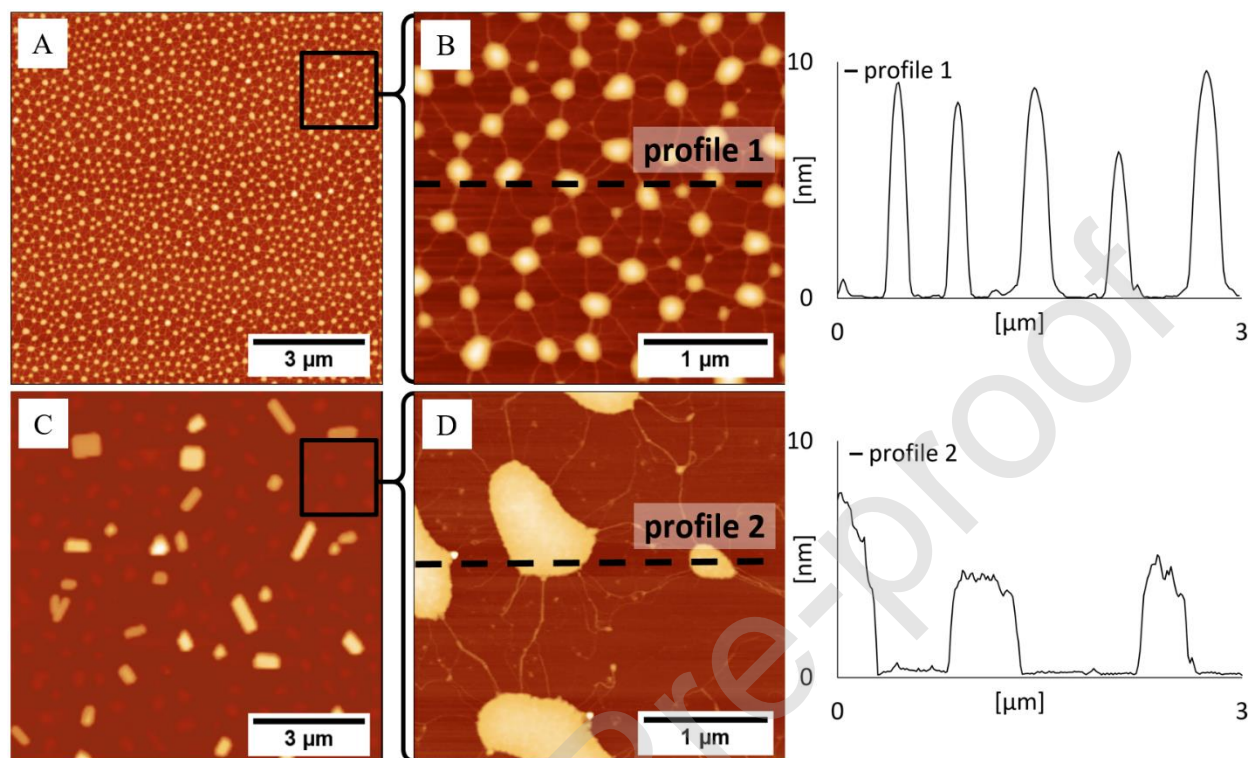
Attributes of HU networks after deposition of the solutions from different dissolving processes, analyzed area  $1 \mu\text{m}^2$

Dissolution process	Aggregates number	Maximum height [nm]	Mean height [nm]	Average projected area $[10^3 \text{ nm}^2]$	Aggregates covering [%]	Network covering [%]	Total covering [%]
Orbital shaker	$8.6 \pm 0.7$	$13.6 \pm 0.6$	$5.5 \pm 0.2$	$22 \pm 1$	$18.3 \pm 1.2$	$25.5 \pm 1.0$	$43.8 \pm 1.6$
Magnetic stirrer	$11.6 \pm 0.8$	$18.7 \pm 0.5$	$8.1 \pm 0.3$	$18 \pm 2$	$19.5 \pm 0.3$	$12.6 \pm 0.6$	$32.0 \pm 0.5$
TFFC device	$13.8 \pm 0.7$	$9.4 \pm 0.5$	$4.6 \pm 0.2$	$12 \pm 1$	$19.3 \pm 0.6$	$39.0 \pm 3.0$	$58.0 \pm 3.0$

### 3.6. HU at physiological condition

In addition to the physical influences on the aggregation and networking of the HU molecules during the solubilization, the influence of increased ionic strength close to physiological conditions was tested. Fig. 7 shows the difference between the structure of HU deposited from water (Fig. 7A) and isotonic 0.9% sodium chloride (Fig. 7B). In the physiological solution, a network structure is formed which is initiated by the formation of hydrogen bridges between the chains, the concentration of ions in the solution has a secondary role in this case (Wu, Ai, Chen, Kang & Cui, 2013). However, larger aggregates of HU on the surface are formed in the presence of saline at the expense of mesh density. In general, the presence of salts promotes a more collapsed HU conformation. We assume that the interaction of these primary clusters leads to the

formation of larger aggregates, which corresponds to previous observations relate to low  $M_w$  HU and long-term storage.



**Fig. 7.** AFM topography images of networks deposited from solution of HU720 in water:

A) surface area  $10 \times 10 \mu\text{m}^2$  and B) surface area  $3 \times 3 \mu\text{m}^2$ ,  $S_z = 13 \text{ nm}$  for both, and in saline: C) surface with NaCl crystals and HU, area  $10 \times 10 \mu\text{m}^2$ ,  $S_z = 400 \text{ nm}$  and D) HU aggregates and chains, surface area  $3 \times 3 \mu\text{m}^2$ ,  $S_z = 12 \text{ nm}$ . HU was dissolved at 0.5 wt. % at room temperature on orbital shaker for 24 h and deposited at concentration of 0.1 wt. %.  $S_z$  means the maximum height of the analyzed area. On the right, the cross sections along the dashed lines are shown.

Polysaccharide properties in solutions are determined mainly by the inter- and intra-chain interactions via hydrogen bonds, hydrophobic interactions and, in the case of polyelectrolytes, by electrostatic interactions. In addition to the intrinsic character of the polymeric chain, a number of extrinsic variables have to be considered in processing of polysaccharide-based systems, Thus,

temperature, ionic strength and pH play also important roles in the conformation changes (Rinaudo, 2006). Further, the pH of the environment also affects the adhesion of HU to the surfaces (Choi et al., 2015), as well as the charge and the chemical composition of the surface (Spagnoli et al., 2005). Specific changes in mesh structure initiated by molecular weight of HU correspond to the well-known HU- $M_w$  dependence, when the intermolecular interactions between HA chains are favoured by chain length (Shen et al., 2011). This phenomenon can have a significant effect on the bioactivity in the preparation of biomaterial surfaces based on hyaluronan (Morra, 2005).

#### **4. Conclusion**

In this study, the network-forming and aggregation properties of HU on the mica surface have been examined using AFM. HU, dissolved at various temperatures and time regimes, were deposited on the charged mica surface and the network structures formed on the liquid-solid interface were imaged in the tapping mode.

The research confirmed the effect of the average molecular weight and the concentration of HU in solution on the aggregation and network-forming properties. In agreement with previous observations, the increase in the mesh density with increasing molecular weight was observed. Furthermore, it has been found that the tendency for crosslinking and aggregation varies during storage of the solution and also depends significantly on its treatment during dissolution, particularly on the temperature and the flow character. The time for which the HU solution is exposed to the dissolution conditions also plays an important role. The observations shows that during storage of water solutions at low temperature, the polymer clusters reorganize and form larger units, similar to the influence of ions present in solution. This rearrangement also occurs

spontaneously in low  $M_w$  HU water solutions. The opposite phenomenon can be observed in case of gentle shear stress of polymer coils in solution.

These findings point to the significant temperature and time dependence of the network-forming and aggregation abilities of HU proving the influence of ambient conditions on the ratio between inter-chain and intra-chain interactions. This knowledge is contributive for progress in understanding the conformational stability of hyaluronan in water solution. Furthermore, presented preparation approaches, especially dissolution HU in temperature gradient, provide a possible way to modify the network texture of HU. Our findings show that in addition to above mentioned physico-chemical influences, the type of applied shear forces generated by the different type of mixing such as orbital, overhead or convectional dissolution process affect the aggregation and crosslinking of HU. In comparison to conventional mixing, apparently increased shear stress in case of the Bénard-Marangoni convection generated by the temperature gradient, enables significant unfolding of the HU chains and inter-crosslinking of the HU network. These finding points towards importance of the shear forces applied during the processing of HU solutions. Systematic studies on the HU solution rheology and its effects on the morphology of the HU network during deposition to investigate systematically the effects of solution viscosity and shear stress are underway to find out new ways for hyaluronan crosslinking in the absence of chemical agents, which is highly desirable in the preparation of hydrogels (Trombino, Servidio, Curcio & Cassano, 2019) with shear thinning behavior suitable for an 3D-printing deposition.

AUTHOR INFORMATION

**Corresponding Author**



Antonín Minařík – *Department of Physics and Materials Engineering, Tomas Bata University in Zlín, Vavrečkova 275, 760 01 Zlín, Czech Republic*

ORCID: 0000-0002-0055-675X; E-mail: [minarik@utb.cz](mailto:minarik@utb.cz)

### **Credit author statement**

Karolína Kocourková: Investigation, Formal analysis, Visualization, Writing – Original draft

Lenka Musilová: Formal analysis

Petr Smolka: Writing – Reviewing and Editing

Aleš Mráček: Project administration, Recourses

Martin Humeník: Investigation, Formal analysis Writing – Reviewing and Editing

Antonín Minařík: Conceptualization, Methodology, Supervision, Writing – Reviewing and Editing

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None.

### **Credit author statement**

## **Factors Determining Self-Assembly of Hyaluronan**

Karolína Kocourková,<sup>†,‡</sup> Lenka Musilová,<sup>†,‡</sup> Petr Smolka,<sup>†,‡</sup> Aleš Mráček,<sup>†,‡</sup> Martin Humeník,<sup>§</sup>

Antonín Minařík,<sup>\*,†,‡</sup>

† Department of Physics and Materials Engineering, Tomas Bata University in Zlín, Vavrečkova 275, 760 01 Zlín, Czech Republic

‡ Centre of Polymer Systems, Tomas Bata University in Zlín, Třída Tomáš Bati 5678, 760 01 Zlín, Czech Republic

§ Department of Biomaterials, Faculty of Engineering Science, Universität Bayreuth, Prof.-Rüdiger-Bormann.Str. 1, 95447 Bayreuth, Germany

### **Credit author statement**

Karolína Kocourková: Investigation, Formal analysis, Visualization, Writing – Original draft

Lenka Musilová: Formal analysis

Petr Smolka: Writing – Reviewing and Editing

Aleš Mráček: Project administration, Recourses

Martin Humeník: Investigation, Formal analysis Writing – Reviewing and Editing

Antonín Minařík: Conceptualization, Methodology, Supervision, Writing – Reviewing and Editing

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